

Lithium Storage

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TiO₂ Hollow Spheres Composed of Highly Crystalline Nanocrystals Exhibit Superior Lithium Storage Properties**

Gengiang Zhang, Hao Bin Wu, Taeseup Song, Ungyu Paik, and Xiong Wen (David) Lou*

Abstract: While the synthesis of TiO₂ hollow structures is well-established, in most cases it is particularly difficult to control the crystallization of TiO₂ in solution or by calcination. As a result, TiO₂ hollow structures do not really exhibit enhanced lithium storage properties. Herein, we report a simple and cost-effective template-assisted method to synthesize anatase TiO₂ hollow spheres composed of highly crystalline nanocrystals, in which carbonaceous (C) spheres are chosen as the removable template. The release of gaseous species from the combustion of C spheres may inhibit the growth of TiO₂ crystallites so that instead small TiO₂ nanocrystals are generated. The small size and high crystallinity of primary TiO₂ nanoparticles and the high structural integrity of the hollow spheres gives rise to significant improvements in the cycling stability and rate performance of the TiO₂ hollow spheres.

Hollow structures with low density, high surface area, and shell permeability have attracted increasing research interest because of their intriguing properties and widespread applications in various fields, including catalysis, drug delivery, sensors, energy storage, and others.^[1-8] Researchers have developed various strategies to controllably synthesize different hollow structures for many materials. [2,9-16] Among these techniques, the hard-templating method, using polymers, silica, and other colloid particles as removable templates, has been considered to be one of the most versatile and straightforward strategies towards hollow spheres. [1,2,17,18] Hollow spheres of various materials, such as metals, metal oxides, and metal sulfides, have been prepared to date using hard-templating methods. For example, carbon-coated SnO₂ hollow spheres have been prepared through a designed multistep synthesis using silica spheres as the template.^[19]

[*] Dr. G. Q. Zhang, Prof. X. W. Lou TUM CREATE, 1 CREATE Way, #10-02 CREATE Tower Singapore 138602 (Singapore)
Dr. G. Q. Zhang, H. R. Wu, Prof. X. W. Lou

Dr. G. Q. Zhang, H. B. Wu, Prof. X. W. Lou School of Chemical and Biomedical Engineering Nanyang Technological University 62 Nanyang Drive, Singapore 637459 (Singapore) E-mail: xwlou@ntu.edu.sg

davidlou88@gmail.com

Homepage: http://www.ntu.edu.sg/home/xwlou

Dr. T. Song, Prof. U. Paik

WCU Department of Energy Engineering, Hanyang University Seoul 133-791 (Korea)

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Despite many successes, some challenging issues remain in conventional hard-templating methods: a) prefunctionalization of the template surface or precise control of the deposition process is usually necessary to obtain uniform coating, b) the process is tedious because of the need to remove templates such as silica spheres, and c) partial collapse of the hollow structures commonly occurs after template removal. As a result, the facile synthesis of high-quality TiO₂ hollow spheres composed of highly crystalline nanocrystals is seldom reported, despite the fact that numerous TiO₂ hollow nanostructures have been reported by different templating or template-free methods. It is thus highly desirable and technologically important to develop simple and scalable strategies for the synthesis of high-quality TiO₂ hollow spheres.

The use of novel electrode materials is one of the central tasks in building the next generation of lithium-ion batteries (LIBs) with high power and energy densities for upcoming large-scale applications. [20-25] Anatase TiO₂ has been long studied as a promising anode material for LIBs because of its unique features, including low cost, environmental benignity, improved safety, and stability. [26-30] However, the low ionic and electrical conductivity which will cause limited capacity and poor cycling performance have seriously hindered its practical electrochemical performance.[31-33] It has been demonstrated that nanosized TiO2 particles could effectively improve the Li⁺ ion diffusion. [29,30] Unfortunately, severe aggregation of nanoparticles during the charge-discharge process imposes another challenge in practical applications. Recently, it has been proposed that TiO₂ hollow structures consisting of nanosized building blocks could partly overcome this problem while retaining high electrochemical activity. [28,31,32,34] However, most of the reported TiO2 hollow structures exhibit less satisfactory long-term cycling stability, which is perhaps as a result of the low crystallinity and relatively large size of nanocrystallites and the poor structural stability of the hollow structures.

Herein, we report a simple and cost-effective template-assisted method to synthesize TiO₂ hollow spheres composed of highly crystalline nanocrystals. Carbonaceous (C) spheres are chosen as the removable template in view of several merits. C spheres can be readily obtained in large quantity from soluble carbohydrates such as glucose through a low-cost and environmentally friendly hydrothermal method. Importantly, the as-prepared C spheres possess rich hydrophilic functional groups, which are essential for their dispersion in solvents and the subsequent uniform deposition amorphous TiO₂ layer. Additionally, C spheres can be easily removed during calcination in air, which produces a highly crystalline TiO₂ product. It is possible that the release of

Figure 1. Illustration of the formation of TiO_2 hollow spheres using carbon spheres as templates: a) amorphous TiO_2 coating (blue) to form the C@ TiO_2 core/shell structure; b) thermal annealing treatment to form polycrystalline TiO_2 hollow spheres (yellow).

gaseous species from combustion of C spheres inhibits the growth of TiO₂ crystallites so that instead small TiO₂ nanocrystals are generated. Together with a proper deposition method, high-quality TiO₂ hollow spheres with a polycrystalline shell and excellent robustness are successfully prepared. The small size and high crystallinity of primary TiO₂ nanoparticles and the high structural integrity of the hollow spheres gives rise to significant improvements in the cycling stability and rate performance of the TiO₂ hollow spheres.

The synthesis of TiO2 hollow spheres is simple and straightforward, as illustrated in Figure 1. The C spheres obtained by a hydrothermal method^[35] were well dispersed into anhydrous ethanol through sonication, followed by the addition of titanium tetrabutoxide (TTB) as the Ti precursor. Next, ammonia solution is added to the reaction mixture to initiate hydrolysis of TTB^[36] in order to obtain a conformal and firm coating of an amorphous TiO₂ layer on the surface of the C spheres (Figure 1, step a). A uniform and stable C@TiO₂ core/shell structure is formed after the reaction, the formation of which is aided by the presence of many hydrophilic functional groups on the surface of the C spheres. Finally, polycrystalline TiO2 hollow spheres composed of small nanocrystals can be obtained after a simple thermal annealing treatment in air (Figure 1, step b). Importantly, this synthesis does not require delicate control of the sol-gel chemistry for the deposition of amorphous TiO₂ on C spheres. In many systems, formation of a uniform shell of sol-gelderived amorphous TiO₂ on templates (for example silica spheres) has been shown to be nontrivial.[37-39]

The morphological characterizations of the C@TiO2 core/ shell intermediate and crystalline TiO2 hollow spheres are shown in Figure 2. Both the field emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) images show uniform C@TiO₂ core/shell spheres prepared through the controlled hydrolysis of TTB on the surface of C spheres (Figures 2A, B). The amorphousness of the C@TiO₂ spheres is confirmed by selected-area electron diffraction (SAED) analysis (see Figure S1 in the Supporting Information). The diameter of these core/shell spheres is in the range of 600 to 800 nm and the variation of diameter is mainly because of the size distribution of the C spheres. After thermal annealing treatment, the spherical morphology is perfectly retained (Figure 2C), and the surface of the spheres is composed of nanoparticles as shown in the magnified FESEM image (Figure 2D). The hollow center of the annealed product is clearly shown through the TEM images (Figures 2 E, F). Unlike the C@TiO₂ core/shell structure, a spherical interior with a distinct contrast difference to the

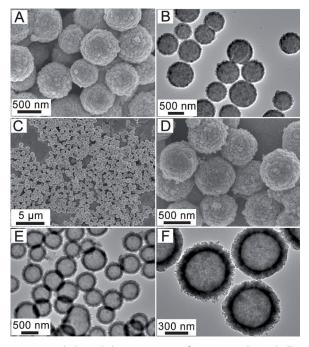


Figure 2. Morphological characterization of $C@TiO_2$ and TiO_2 hollow spheres: A) FESEM and B) TEM images of $C@TiO_2$ core/shell spheres; C, D) FESEM and E, F) TEM images of crystallized TiO_2 hollow spheres under different magnifications.

shell is evident in the center of the annealed spheres. The shell thickness is estimated to be about 100 nm with a relatively rough outer surface. The typical diameter of TiO₂ hollow spheres decreases slightly to about 500–700 nm. This could be ascribed to a contraction effect caused by the decomposition of C spheres at elevated temperature. The phase purity of the annealed sample is examined by powder X-ray diffraction (XRD) analysis (Figure S2), where the diffraction pattern can be readily indexed to the anatase TiO₂ phase (JCPDS card no. 21-1272). The broadened diffraction peaks suggest that the nanocrystals are small in size. According to the Scherrer formula, the average crystallite size is calculated to be approximately 9.1 nm.

The microstructure of TiO2 hollow spheres is further analyzed by high-resolution TEM (HRTEM), as shown in Figure 3. Lattice fringes are clearly evident from both the center (Figure 3B) and the edge (Figure 3C) of a randomly selected TiO₂ hollow sphere shown in Figure 3A. These lattice fringes indicate that the product is highly crystalline after thermal annealing. The interplanar distance between the lattice fringes is measured to be 0.35 nm, which can be indexed to (101) crystal planes of anatase TiO₂. Additionally, there are notable grain boundaries from the lattice fringes shown in Figures 3B and C, which indicate that the shell of the hollow spheres is composed of small nanocrystals. The size of the nanocrystals is estimated to be approximately 10 nm, which is in good agreement with the value calculated from XRD analysis. These TiO₂ hollow spheres show a relatively high Brunauer-Emmett-Teller (BET) specific surface area of 71.2 m² g⁻¹ (N₂ adsorption-desorption isotherms are given in Figure S4). The SAED pattern of TiO₂ hollow spheres



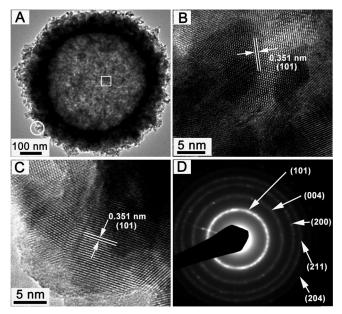


Figure 3. A) TEM image of an individual TiO_2 hollow sphere upon which the microstructure analysis is performed; B) HRTEM image taken from the part of the hollow sphere indicated with a square in (A); C) HRTEM image taken from the part of the hollow sphere indicated with a circle in (A); D) corresponding SAED pattern.

(Figure 3D) again indicates the polycrystalline nature of the sample and each of the diffraction rings can be readily indexed to anatase TiO₂, which is consistent with the XRD results. Moreover, no visible gap can be detected between the nanocrystals, which implies that the hollow spheres have good mechanical integrity. The high robustness of these hollow spheres is further verified by sonicating the sample for 2 hours (Figure S3). From the TEM images, no notable variation in morphology or collapse of the hollow structure is evident. This excellent structural stability would be favorable for reversible lithium storage with prolonged cycle life.

To demonstrate the advantages of these robust TiO₂ hollow spheres, we have evaluated their lithium storage properties as anode materials for LIBs. The cyclic voltammograms (Figure S5) exhibit the characteristic lithium insertion/ de-insertion behavior for anatase TiO2, with two redox peaks recorded at approximately 1.6 V and 2.3 V, respectively, versus Li⁺/Li. Figure 4 A shows representative dischargecharge voltage profiles within a cut-off voltage window of 1.0– 3.0 V versus Li⁺/Li. There are two notable voltage plateaus in the discharge-charge curves at approximately 1.7 V and 2.0 V versus Li⁺/Li, respectively, which correspond to the lithium insertion/de-insertion process.^[28] The initial discharge and charge capacities are 187.4 and 148.6 mA h g⁻¹, respectively, leading to a high coulombic efficiency of 79.3 %. Additionally, discharge and charge curves of the second and the hundredth cycles are almost identical, indicating that the electrochemical process is stable during the lithium insertion/de-insertion reactions. Figure 4B shows the cycling performance of TiO₂ hollow spheres at a current density of 1 C (1 C = 173 mA g^{-1}). The capacity decays from 187.4 to 147.6 mA h g⁻¹ in the first 10 cycles and remains very stable up to 300 cycles. The longterm cycling stability of our TiO₂ hollow spheres is superior to

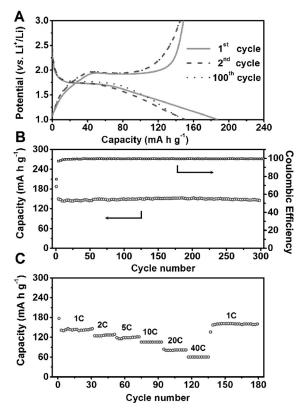


Figure 4. Electrochemical performance of TiO_2 hollow spheres as an anode material in LIBs: A) discharge–charge voltage profiles in the voltage range of 1.0–3.0 V; B) cycling performance and corresponding Coulombic efficiency at a current rate of 1 C; C) rate performance at various current rates from 1 C to 40 C. 1 C = 173 mAg⁻¹.

that of other TiO₂ materials reported. [28,31,32] Moreover, these TiO₂ hollow spheres exhibit excellent rate capability at discharge-charge current rates ranging from 1 to 40 C, as shown in Figure 4C. The average specific capacities are 145.2, 127.3, 120.1, 105.6, 81.4, and 60.8 mA h g^{-1} at current rates of 1, 2, 5, 10, 20, and 40 C, respectively. After the high-rate discharge-charge cycling, a specific capacity of 158.2 mA h g⁻¹ can be restored when the current density is decreased to 1 C. Moreover, a study of the material after all cycling experiments reveals that the hollow structure is perfectly retained after cycling at 5 C for 100 cycles (Figure S6), indicating the excellent structural robustness of these TiO₂ hollow spheres. These results clearly demonstrate the superior lithium storage properties of TiO₂ hollow spheres in terms of long cycle life and a good rate capability for the fast charging/discharging process.

The outstanding electrochemical performance of the as-prepared ${\rm TiO_2}$ hollow spheres as anode materials for LIBs can be understood from several perspectives. First, the small size of the primary nanocrystals and the relatively thin shell offer a short diffusion distance for Li⁺ions, thus promoting fast and reversible lithium insertion and extraction. Second, the robust shell structure with a sub-micrometer size effectively prevents the undesirable aggregation of conventional nanoparticles, which ensures the integrity of the electrode and improves the capacity retention upon

prolonged cycling. Third, the hollow structure with high surface area provides more active surface sites and electrolyte-electrode interface compared with solid counterparts. Additionally, the high crystallinity of TiO2 hollow spheres may contribute to the electrical conductivity and crystal lattice robustness during repeated charge-discharge cycling. These features would promote the electrochemical process and lead to high specific capacity especially at high rates.

In summary, anatase TiO₂ hollow spheres composed of highly crystalline small nanocrystals are successfully fabricated through a simple templating approach using carbonaceous spheres as hard templates. The present method is facile and suitable for low-cost large-scale production of TiO2 hollow spheres. Owing to the robust hollow structure and small primary nanoparticles, these TiO2 hollow spheres exhibit exceptional lithium storage properties with ultrastable capacity retention for over 300 cycles and excellent rate capability up to 40 C. This work is expected to be useful for the development of high-performance TiO₂-based anodes for lithium-ion batteries.

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- [1] F. Caruso, R. A. Caruso, H. Mohwald, Science 1998, 282, 1111-1114.
- [2] X. W. Lou, L. A. Archer, Z. C. Yang, Adv. Mater. 2008, 20, 3987 -
- [3] J. Hu, M. Chen, X. S. Fang, L. W. Wu, Chem. Soc. Rev. 2011, 40, 5472 – 5491.
- [4] X. Y. Lai, J. E. Halpert, D. Wang, Energy Environ. Sci. 2012, 5, 5604 - 5618.
- [5] J. B. Joo, Q. Zhang, M. Dahl, I. Lee, J. Goebl, F. Zaera, Y. D. Yin, Energy Environ. Sci. 2012, 5, 6321-6327.
- [6] Y. Piao, J. Kim, H. Bin Na, D. Kim, J. S. Baek, M. K. Ko, J. H. Lee, M. Shokouhimehr, T. Hyeon, Nat. Mater. 2008, 7, 242 – 247.
- [7] Y. Zhao, L. Jiang, Adv. Mater. 2009, 21, 3621-3638.
- [8] X. Wang, W. Tian, T. Zhai, C. Zhi, Y. Bando, D. Golberg, J. Mater. Chem. 2012, 22, 23310-23326.
- [9] J. Wang, N. Yang, H. Tang, Z. Dong, Q. Jin, M. Yang, D. Kisailus, H. Zhao, Z. Tang, D. Wang, Angew. Chem. 2013, 125, 6545-6548; Angew. Chem. Int. Ed. 2013, 52, 6417-6420.
- [10] Y. J. Hong, M. Y. Son, Y. C. Kang, Adv. Mater. 2013, 25, 2279-
- [11] G. Q. Zhang, X. W. Lou, Angew. Chem. Int. Ed. 2014, DOI: 10.1002/anie.201404604; Angew. Chem. 2014, DOI: 10.1002/ ange.201404604.

- [12] G. Zhang, L. Yu, H. B. Wu, H. E. Hoster, X. W. Lou, Adv. Mater. **2012**, 24, 4609-4613.
- [13] L. Hu, H. Zhong, X. R. Zheng, Y. M. Huang, P. Zhang, Q. W. Chen, Sci. Rep. 2012, 2, 986.
- [14] Z. W. Seh, W. Y. Li, J. J. Cha, G. Y. Zheng, Y. Yang, M. T. McDowell, P. C. Hsu, Y. Cui, Nat. Commun. 2013, 4, 1331.
- [15] M. H. Oh, T. Yu, S.-H. Yu, B. Lim, K.-T. Ko, M.-G. Willinger, D.-H. Seo, B. H. Kim, M. G. Cho, J.-H. Park, K. Kang, Y.-E. Sung, N. Pinna, T. Hyeon, Science 2013, 340, 964-968.
- [16] Y. D. Yin, R. M. Rioux, C. K. Erdonmez, S. Hughes, G. A. Somorjai, A. P. Alivisatos, Science 2004, 304, 711-714.
- [17] X. Sun, J. Liu, Y. Li, Chem. Eur. J. 2006, 12, 2039-2047.
- [18] Z. H. Dong, X. Y. Lai, J. E. Halpert, N. L. Yang, L. X. Yi, J. Zhai, D. Wang, Z. Y. Tang, L. Jiang, Adv. Mater. 2012, 24, 1046 – 1049.
- [19] X. W. Lou, C. M. Li, L. A. Archer, Adv. Mater. 2009, 21, 2536-2539.
- [20] P. G. Bruce, B. Scrosati, J.-M. Tarascon, Angew. Chem. 2008, 120, 2972-2989; Angew. Chem. Int. Ed. 2008, 47, 2930-2946.
- [21] J. Cabana, L. Monconduit, D. Larcher, M. R. Palacin, Adv. Mater. 2010, 22, E170-E192.
- [22] Y.-G. Guo, J.-S. Hu, L.-J. Wan, Adv. Mater. 2008, 20, 2878 2887.
- [23] N.-S. Choi, Z. Chen, S. A. Freunberger, X. Ji, Y.-K. Sun, K. Amine, G. Yushin, L. F. Nazar, J. Cho, P. G. Bruce, Angew. Chem. 2012, 124, 10134-10166; Angew. Chem. Int. Ed. 2012, 51, 9994 - 10024.
- [24] R. Mukherjee, R. Krishnan, T.-M. Lu, N. Koratkar, Nano Energy **2012**, 1, 518-533.
- [25] A. S. Aricò, P. Bruce, B. Scrosati, J.-M. Tarascon, W. Schalkwijk, Nat. Mater. 2005, 4, 366-377.
- [26] J. S. Chen, Y. L. Tan, C. M. Li, Y. L. Cheah, D. Y. Luan, S. Madhavi, F. Y. C. Boey, L. A. Archer, X. W. Lou, J. Am. Chem. Soc. 2010, 132, 6124-6130.
- [27] X. W. Lou, L. A. Archer, Adv. Mater. 2008, 20, 1853-1858.
- [28] Z. Y. Wang, X. W. Lou, Adv. Mater. 2012, 24, 4124-4129.
- [29] S. T. Myung, N. Takahashi, S. Komaba, C. S. Yoon, Y. K. Sun, K. Amine, H. Yashiro, Adv. Funct. Mater. 2011, 21, 3231-3241.
- [30] X. Chen, S. S. Mao, Chem. Rev. 2007, 107, 2891 2959.
- [31] J. P. Wang, Y. Bai, M. Y. Wu, J. Yin, W. F. Zhang, J. Power Sources **2009**, 191, 614–618.
- [32] L. Xiao, M. L. Cao, D. D. Mei, Y. L. Guo, L. F. Yao, D. Y. Qu, B. H. Deng, J. Power Sources 2013, 238, 197-202.
- [33] F. Gligor, S. W. de Leeuw, Solid State Ionics 2006, 177, 2741 -
- [34] S. J. Ding, T. Q. Lin, Y. M. Wang, X. J. Lu, F. Q. Huang, New J. Chem. 2013, 37, 784-789.
- [35] X. M. Sun, Y. D. Li, Angew. Chem. 2004, 116, 607–611; Angew. Chem. Int. Ed. 2004, 43, 597-601.
- [36] W. Li, J. P. Yang, Z. X. Wu, J. X. Wang, B. Li, S. S. Feng, Y. H. Deng, F. Zhang, D. Y. Zhao, J. Am. Chem. Soc. 2012, 134, 11864-
- [37] L. Yu, H. B. Wu, X. W. Lou, Adv. Mater. 2013, 25, 2296-2300.
- [38] J. B. Joo, I. Lee, M. Dahl, G. D. Moon, F. Zaera, Y. Yin, Adv. Funct. Mater. 2013, 23, 4246-4254.
- [39] Z. Yang, Z. Niu, Y. Lu, Z. Hu, C. C. Han, Angew. Chem. 2003, 115, 1987-1989; Angew. Chem. Int. Ed. 2003, 42, 1943-1945.

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